

基于振动时效仿真的流变塑性模型

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摘要: 振动时效是一种局部循环塑性现象, 当循环载荷与材料内部残余应力叠加超过材料的局部屈服强度时就会发生残余应力释放。采用流变塑性模型对振动时效进行仿真, 分析了振动时效过程中应力幅、应变幅、振动频率、振动周期和材料屈服应力等对振动时效的影响。结果表明, 振动时效中应力释放很大程度上取决于应力幅或应变幅, 振动频率和材料参数也是关键因素, 而振动周期或时间对振动时效没有很大影响。将此振动时效模型应用于 7075 铝合金试样机械载荷下的应力松弛实验, 所得结果与仿真较一致。

关键词: 流变塑性; 应力幅; 应变幅; 残余应力; 峰值应力

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Simulation of Vibratory Stress Relief Based on Rheological Plasticity Model

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Abstract: Vibratory stress relief (VSR) is a localized cyclic plasticity phenomenon, and residual stress relief occurs when the vector superposition of induced cyclic stress and residual stress exceeds the local yield stress of the material. Simulations of Vibratory stress relief were processed by a rheological plasticity model, in which the effects of dynamic stress amplitude, dynamic strain amplitude, vibration frequency, vibration cycle and material yield stress on VSR were analyzed. The results show that residual stress relief during VSR depends mainly on stress amplitude or strain amplitude. In addition, vibration frequency and material parameters are also key factors while vibration cycle or time has no great effects on VSR. Simulations of the rheological plasticity model have a good agreement with experimental results from 7075 aluminum alloy specimens, which could be instructive and meaningful to experiments and engineering applications.

Key words: rheological plasticity; stress amplitude; strain amplitude; residual stress; peak stress

1 前言

在大中型金属构件的制作过程中, 诸如焊接、热处理、铸造、机加工等生产过程通常会对零部件或结构引入残余应力, 给构件的力学性能与尺寸稳定性带来严重影响^[1-3]。

近年来发展的振动时效技术(简称 VSR)因其诸多优点发展迅速, 应用已经十分广泛, 然而由于缺少对振动消减残余应力这种现象的透彻理解, 所以在相当程度上制约了这项技术的发展^[4-8]。VSR 理论在阐释降低和

均匀化构件的残余应力的原理时, 经常使用的最基本公式为^[9-11]:

$$\begin{cases} \sigma_d + \sigma_R > \sigma_y \\ \sigma_d = \sigma_a + \sigma_m \end{cases} \quad (1)$$

式中, σ_d 为振动时激振器提供的动应力, σ_a 与 σ_m 分别为其应力幅与平均应力分量; σ_R 为构件的残余应力; σ_y 为材料的屈服极限。一般认为, 振动时构件的某些区、段或点上, σ_d 和 σ_R 叠加, 若使式(1)成立, 则由于发生微观塑性变形而会使原有的残余应力释放。基于这一理论, 本文采用流变塑性模型对振动时效仿真的典型响应, 主要分析振动时效参数, 如振动频率、应力幅、应变幅和振动时间等对振动时效效果的影响, 以便了解振动时效过程的机理并对振动时效过程的参数进行优化。模型分析结果与 7075 铝合金试样应力松弛实验结果一致, 说明将此塑性模型应用于振动时效是较为准

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2 模型建模与仿真

振动时效时零部件材料对外加动载荷的机械响应主要取决于施加的动应力、频率和系统的阻尼性能。在一个循环中，残余应力表现为平均应力，循环应力和残余应力的交互作用将导致偏应力循环载荷条件，可认为是残余应力释放的驱动力。

材料对振动时效处理响应的整体循环弹塑性特性可表示为^[12-13]：

$$\frac{d\sigma}{d\varepsilon} = E, \sigma\left(\frac{d\sigma}{dt}\right) < 0 \text{ 时} \quad (2a)$$

$$\frac{d\sigma}{d\varepsilon} = \theta_d \left(1 - \frac{\lambda\sigma}{\sigma_{pk}}\right), \sigma\left(\frac{d\sigma}{dt}\right) \geq 0 \text{ 且 } \sigma \geq 0 \text{ 时} \quad (2b)$$

$$\frac{d\sigma}{d\varepsilon} = \theta_d \left(1 - \frac{\lambda\sigma}{\sigma_{pc}}\right), \sigma\left(\frac{d\sigma}{dt}\right) \geq 0 \text{ 且 } \sigma < 0 \text{ 时} \quad (2c)$$

$$\frac{d\sigma_{pk}}{dN} = 4\varepsilon_a\theta_0 \left(1 - \frac{\sigma_{pk}}{\sigma_{pks}}\right) \quad (2d)$$

$$\sigma_{pk} = \sigma_a + \sigma_R \quad (2e)$$

$$\varepsilon = \varepsilon_a \cos(\omega t) + \varepsilon_m \quad (2f)$$

式中， θ_0 与 θ_d 为流动应力-应变曲线的初始斜率； λ 为常数； σ_{pk} 与 σ_{pc} 为峰值拉压应力； σ_{pks} 为残余应力释放停止时的饱和峰值应力； σ_a 为外部激振力的循环应力幅； σ_R 为残余应力； ε_a 为循环应变幅； ε_m 为一个周期内的平均应变； ω 为循环角频率； t 为时间。

加上适当的边界条件，方程(2)可用来描述材料加载、卸载和反向载荷条件下循环塑性应力-应变特性。应力释放中有 $|\sigma_{pk}| + |\sigma_{pc}| = 2\sigma_a$ ，由于应力幅 σ_a 在振动时效过程中保持为常量，这也表明若 σ_{pk} 改变，根据方程(2e)，是由于残余应力 σ_R 得到释放。

仿真模型应力松弛基本参数如表1所示，代入方程(2)并转换为如图1所示的 Matlab/Simulink 块体模型图，得到振动时效下的应力-应变特性与峰值应力随周期变化的应力释放曲线，分别如图2与图3所示。

表1 仿真模型基本参数

Table 1 Parameters of simulation model

σ_a /MPa	σ_{R0} /MPa	σ_y /MPa	σ_{pks} /MPa	θ_d	λ	E /GPa
60	110	110	110	2 800	0.97	120

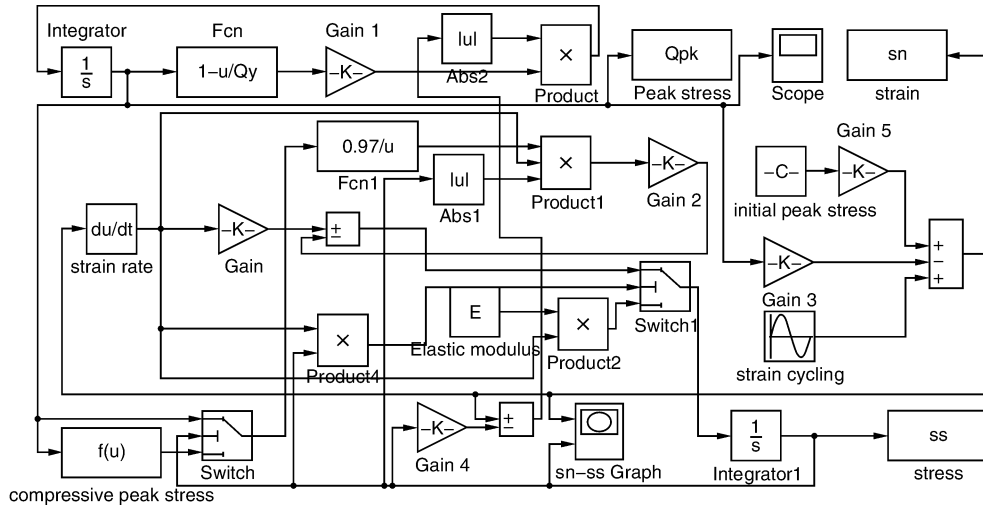


图1 振动时效下机械响应仿真的 Matlab/Simulink 框图

Fig. 1 Matlab/Simulink Block Diagram for simulating mechanical response of material subjected to VSR treatment

图2表明，当峰值拉应力 σ_{pk} 下降至 σ_y 时应力松弛趋于平缓，而当引入的循环应力幅值 σ_a 与残余应力 σ_R 之和超过材料的局部屈服应力时，材料将发生塑性变形，塑性变形导致应力释放，并且在加载、卸载和反向加载时遵循不同的应力-应变路径。理论上，应力释放将会在应力幅与残余应力之和低于材料的局部屈服应力时停止，由图2可知在之后的载荷周期中峰值应力将保持为常数。

图3中用符号表示的离散数据来自于图2中峰值应力，而光滑曲线是由方程(2d)给出，表明方程(2)可以

用来仿真振动时效引起的应力释放。

3 模型参数评估

3.1 动应力幅

振动时效时构件所承受的载荷由构件本身的残余应力和施加在构件上的动应力两部分组成。构件振动的动应力来自于激振装置的激振力，对于机械式激振装置，激振力就是偏心质量的旋转产生的离心力。激振力一般认为是对称循环应力，与残余应力叠加后为不对称循环应力，故构件内各点的应力为不对称循环应力。本质上

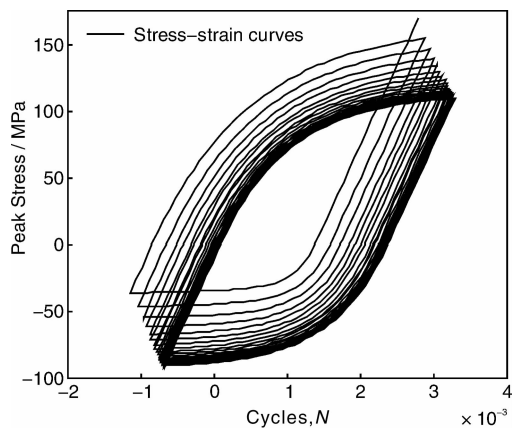


图 2 振动时效下残余应力释放的仿真应力-应变响应

Fig. 2 Simulated stress-strain response under VSR

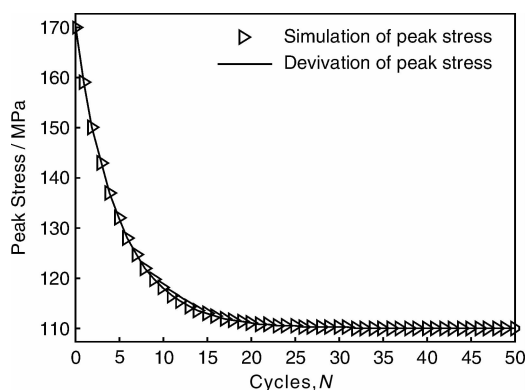


图 3 峰值应力随周期变化的应力释放曲线

Fig. 3 Stress relaxation showing variation of peak stress with cycle

振动时效类似于应力松弛现象, 图 4 为仅改变应力幅时峰值应力随时间或周期的减小。

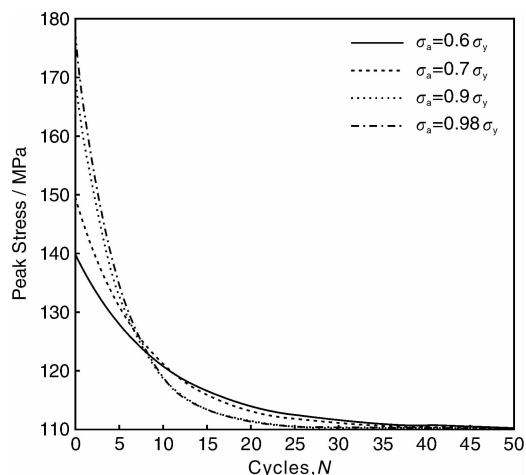


图 4 不同动应力幅下的残余应力松弛

Fig. 4 Residual stress relief under different stress amplitudes

由方程(2e)可知, 图中的 σ_{pk} 减小量等于残余应力

的松弛量。随着动应力幅的增加, 残余应力的松弛量与松弛速率均增大, 当应力幅超过材料屈服极限的一半, 残余应力松弛速率大幅增加。如当 $\sigma_a = 0.6\sigma_y$ 和 $\sigma_a = 0.7\sigma_y$ 时, 前 10 个周期残余应力松弛 30% 左右, 45 个周期残余应力松弛近一半; 而当应力幅接近屈服极限, 如 $\sigma_a = 0.9\sigma_y$ 和 $\sigma_a = 0.98\sigma_y$ 时, 前 5 个周期残余应力松弛一半。方程(1)中的动应力 σ_d 是达到 VSR 工艺目的的最佳可控项, 在实际应用中大多数情况下为施加的动应力幅 σ_a 。根据位错理论, 只要动应力足够大, 超过临界值时, 其它取向与之相近的晶粒内的位错将陆续启动和滑移, 从而使残余应力释放, 动应力越大, 位错就越容易发生和增殖^[14]。

3.2 动应变幅

残余应力释放发生在恒定的应变幅下, 以数学形式表示就是应变幅 ε_a 为弹性 ε_e 和塑性 ε_i 部分之和:

$$\varepsilon_a = \varepsilon_e + \varepsilon_i \quad (3)$$

塑性应变在损害弹性零部件的情况下持续增加, 而弹性应变的降低也表明弹性残余应力得以重新分布。构件在振动时, 材料内应变幅越大则振动时效效果越好, 大的应变幅更容易在微观尺度上启动位错运动, 从而引起应力释放。图 5 所示为不同应变幅下的残余应力释放, 与应力幅对振动时效的影响类似。

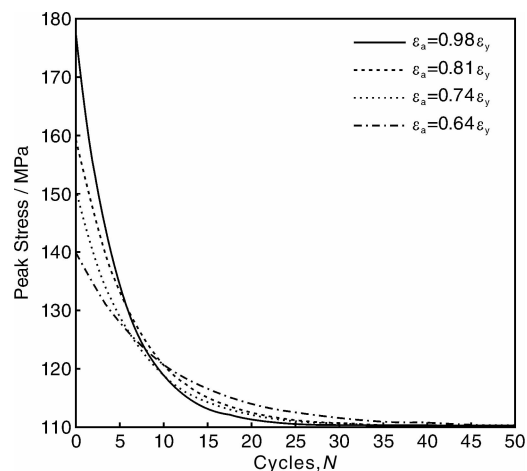


图 5 不同应变幅下的残余应力松弛

Fig. 5 Residual stress relief under different strain amplitudes

3.3 振动频率

图 6 为文中塑性模型仿真出的振动频率对残余应力松弛的影响。频率越高, 构件内部单位时间内得到的能量越大, 累积到促使材料内部发生位错启动及增殖的能量所需时间越少, 故残余应力松弛速率越大。

振动时效系统一般可认为是一个受简谐激励的受迫振动多自由度系统。构件的固有频率与其刚度、质量、

阻尼与边界条件等有关。振动时效工作在共振状态时,波节处剪应变很大,剪应变率最高,材料中部分晶粒在剪切应力作用下发生微观塑性变形,构件上原有的残余应力与剪切动应力或拉压应力的剪切分量共同作用,从而可用最小的振动能量使构件产生最大的振幅和动应力,使构件中的残余应力消除最多,尺寸稳定性效果最好。故一般选取一阶或二阶共振频率作为激振频率,因为低价共振频率振型节点少,频率虽低但容易获得较大振幅。此模型并不能同时模拟振动系统的共振频率,但可以表明一阶共振频率之前随振动频率增加,残余应力松弛速率增加这一事实。

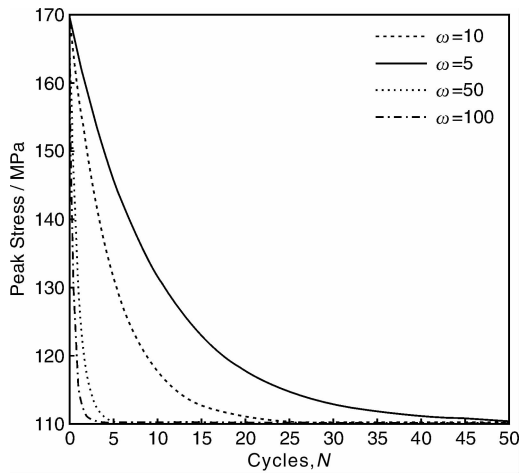


图6 不同振动频率下的残余应力松弛

Fig. 6 Residual stress relief under different frequencies

3.4 振动周期

构件在工作期间,表面的压缩残余应力往往承受平均应力为正的循环载荷。这种情况下,残余应力松弛在初始阶段的疲劳循环中非常显著,在极端情形下残余应力可以在前几个载荷循环中完全释放。如图7所示,对构件原始为压缩的残余应力状态施加大的应力幅,在5~10个周期残余应力将下降至原始应力水平的80%,对于动应力幅接近材料屈服极限时甚至降低至20%左右,30个周期后应力松弛趋于平缓,残余应力释放停止或很少。当应力幅接近材料的屈服极限时,如 $\sigma_a = 0.9\sigma_y$,残余应力在30个周期内松弛速率趋于平缓,松弛量与动应力幅有关。

振动周期或激振时间对振动时效效果影响不大,故振动时效中,低的动应力幅下常对构件激振15~30 min,激振时间过多不仅对残余应力的释放没有作用,而且影响效率,甚至对构件造成永久损坏。

3.5 材料屈服强度

方程(2)描述的振动时效残余应力释放塑性模型预

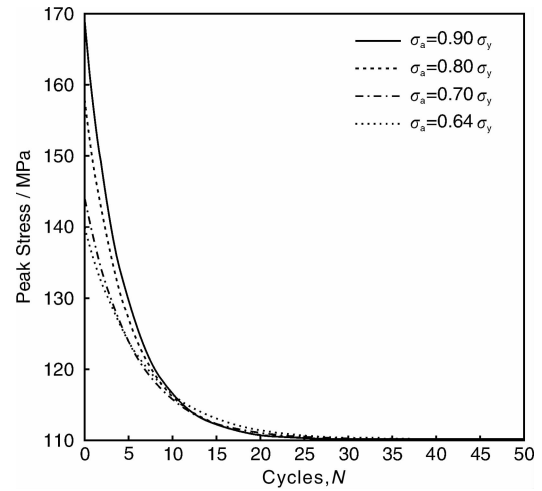


图7 不同振动周期下的残余应力松弛

Fig. 7 Residual stress relief under different vibration period

测出对于给定材料,若采用高应力幅、高应变幅、工作在谐振频率下,则振动时效过程将更为有效。残余应力释放开始时峰值应力超过了局部屈服应力,而此时的饱和应力 σ_{pks} 可认为与材料的局部屈服应力 σ_y 相等,将代入方程(2)可得:

$$\sigma_R = \sigma_{R0} - (\sigma_{pks} - \sigma_y) [1 - \exp(\frac{-4\varepsilon_a N \theta_0}{\sigma_y})] \quad (4)$$

式中, σ_{pks} 为初始峰值应力; N 为周期。式(4)表明,具有高屈服应力或很低的初始应变硬化率,即低延展性和高硬度的材料在振动时效处理时往往效果欠佳。

图8为不同屈服应力对振动时效效果的影响。由图可知,动应力幅或应变幅不变时,材料屈服强度越高或冷加工程度越深,应力释放速率就越慢,且应力释放程度也越低。

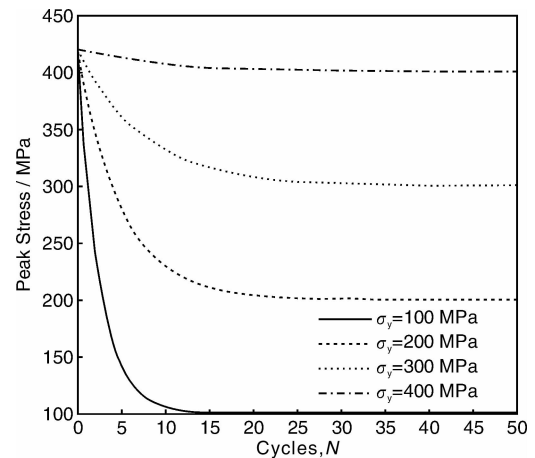


图8 材料屈服强度对振动时效的影响

Fig. 8 Residual stress relief under different material yield stress

4 模型应用

实验材料为 7075 铝合金(主要成分:铜 1.6%, 镁 2.5%, 铬 0.23%, 锌 5.6%, 其余为铝), 7075 铝合金试样尺寸如图 9 所示。

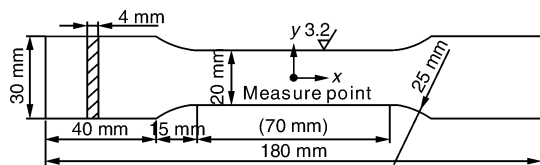


图 9 淬火态 7075 铝合金试样

Fig. 9 Quenched 7075 aluminum specimen

一组试样经退火与表面喷砂引入表层残余压应力, 采用 Proto X 衍射仪测量试样表面中心纵向残余应力值。采用华龙 WPL-250 型动静万能试验机标定试件的力学性能及引入动应力幅(应力比 $R = -1$), 试样单向拉伸的力学性能及表面初始残余应力值列于表 2。

表 2 7075 铝合金试样力学性能

Table 2 Mechanical properties of 7075 aluminum

Specimen	E/GPa	$R_{p0.2}/\text{MPa}$	R_m/MPa	σ_{R0}/MPa
7075	68.3	295	475	-133

实验结果如图 10 所示, 实验结果与仿真较一致。当 $\sigma_a = 0.2\sigma_y$ 和 $\sigma_a = 0.4\sigma_y$ 时应力幅较小, 与残余应力叠加的复合应力低于材料屈服应力, 表面应力几乎没有松弛; 当 $\sigma_a = 0.6\sigma_y$, $\sigma_a = 0.8\sigma_y$ 与 $\sigma_a = \sigma_y$ 左右时, 复合应力大于屈服强度, 试样表面残余应力在第一个周期后应力即明显松弛, 且随复合应力逐渐增加, 试样表面应力下降加快, 前 30 个周期应力松弛基本完全, Rao 等人^[15]通过实验也观察到类似结果。

5 结论

振动时效一般是在室温下进行, 施加的动应力与材料内部残余应力叠加使吉布斯自由能下降从而激活应力释放。振动时效是一种局部非均匀循环塑性现象, 材料不同部分具有不同的循环应力-应变行为, 方程(2)描述的振动时效残余应力释放模型能够仿真多参数的影响, 对实验或工程应用有一定的指导作用。

振动时效过程中施加的动应力幅或应变幅为最重要因素, 高的应力幅或应变幅能够显著改善残余应力的释放; 振动频率对振动时效影响较大, 在系统谐振频率附近激振时, 材料内部的微小应变将被显著放大从而改善残余应力释放; 振动周期或时间对振动时效影响并不显著, 一般情况下残余应力大幅消减发生在前期, 增加振动时间反而会降低构件的疲劳寿命; 材料参数对振动时效也有很大影响, 屈服强度过高及冷加工程度过深的材

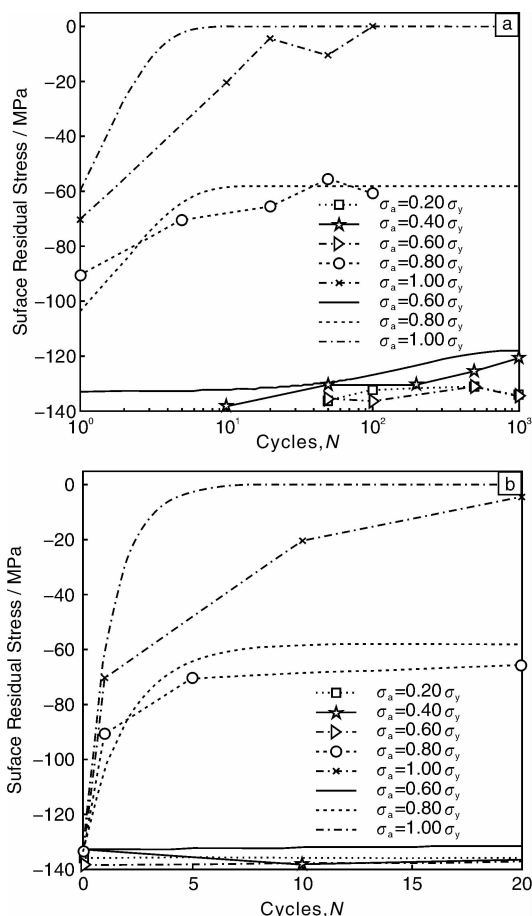


图 10 试样表面残余应力松弛曲线

Fig. 10 Residual stress relief curves of specimen surface:

(a) Cycles 1 ~ 10^3 , and (b) Cycles 0 ~ 20

料振动时效效果欠佳。

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Say Hello to Cheaper Hydrogen Fuel Cells: Scientists Document Utility of Non-Precious-Metal Catalysts

Los Alamos National Laboratory scientists have developed a way to avoid the use of expensive platinum in hydrogen fuel cells, the environmentally friendly devices that might replace current power sources in everything from personal data devices to automobiles.

In a paper published today in *Science*, Los Alamos researchers Gang Wu, Christina Johnston, and Piotr Zelenay, joined by researcher Karen More of Oak Ridge National Laboratory, describe the use of a platinum-free catalyst in the cathode of a hydrogen fuel cell. Eliminating platinum—a precious metal more expensive than gold—would solve a significant economic challenge that has thwarted widespread use of large-scale hydrogen fuel cell systems.

Polymer-electrolyte hydrogen fuel cells convert hydrogen and oxygen into electricity. The cells can be enlarged and combined in series for high-power applications, including automobiles. Under optimal conditions, the hydrogen fuel cell produces water as a “waste” product and does not emit greenhouse gasses. However, because the use of platinum in catalysts is necessary to facilitate the reactions that produce electricity within a fuel cell, widespread use of fuel cells in common applications has been cost prohibitive. An increase in the demand for platinum-based catalysts could drive up the cost of platinum even higher than its current value of nearly \$1,800 an ounce.

The Los Alamos researchers developed non-precious-metal catalysts for the part of the fuel cell that reacts with oxygen. The catalysts—which use carbon (partially derived from polyaniline in a high-temperature process), and inexpensive iron and cobalt instead of platinum—yielded high power output, good efficiency, and promising longevity. The researchers found that fuel cells containing the carbon-iron-cobalt catalyst synthesized by Wu not only generated currents comparable to the output of precious-metal-catalyst fuel cells, but held up favorably when cycled on and off—a condition that can damage inferior catalysts relatively quickly.

Moreover, the carbon-iron-cobalt catalyst fuel cells effectively completed the conversion of hydrogen and oxygen into water, rather than producing large amounts of undesirable hydrogen peroxide. Inefficient conversion of the fuels, which generates hydrogen peroxide, can reduce power output by up to 50 percent, and also has the potential to destroy fuel cell membranes. Fortunately, the carbon-iron-cobalt catalysts synthesized at Los Alamos create extremely small amounts of hydrogen peroxide, even when compared with state-of-the-art platinum-based oxygen-reduction catalysts.

Because of the successful performance of the new catalyst, the Los Alamos researchers have filed a patent for it.

“The encouraging point is that we have found a catalyst with a good durability and life cycle relative to platinum-based catalysts,” said Zelenay, corresponding author for the paper. “For all intents and purposes, this is a zero-cost catalyst in comparison to platinum, so it directly addresses one of the main barriers to hydrogen fuel cells.”

The next step in the team’s research will be to better understand the mechanism underlying the carbon-iron-cobalt catalyst. Micrographic images of portions of the catalyst by researcher More have provided some insight into how it functions, but further work must be done to confirm theories by the research team. Such an understanding could lead to improvements in non-precious-metal catalysts, further increasing their efficiency and lifespan.

(Provided by Los Alamos National Laboratory)